# 铝合金气体输送活性钨极氩弧焊方法

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摘 要:针对铝合金 提出了一种气体输送活性钨极氩弧焊,即 GTFA-TIG 焊(gas transfer flux activating TIG welding). 该方法改变了活性元素的引入方式,通过自动送粉装置将活性剂输送到保护气体中,由保护气体将其引入电弧一熔池系统进行施焊,使得电弧收缩,熔池金属流态改变,熔深增加,同时省却了涂覆活性剂工序,实现了焊接过程自动化。进行了普通交流 TIG 焊和 8 种单组元活性剂的 GTFA-TIG 表面熔焊,分析了不同活性剂对焊缝成形、拉伸性能以及缺陷的影响。结果表明,大多数卤化物和氧化物活性剂都能使熔深增加到传统 TIG 焊的  $2.5 \sim 3$  倍以上,单质碲增加熔深效果较差。采用  $V_2O_5$  的焊缝抗拉强度接近母材金属,而采用  $MnCl_2$  和  $AlF_3$  的焊缝有一定程度降低.焊缝 X 射线探伤结果表明 采用  $V_2O_5$   $MnCl_2$  和  $AlF_3$  的焊缝评片结果均为 I 级,而采用碲的焊缝为 II 级。

关键词: 气体输送活性钨极氩弧焊; 铝合金; 电弧收缩; 活性剂

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## 0 序 言

近些年来 活性焊接法引起了世界范围内的重视<sup>[12]</sup> ,它起初由乌克兰巴顿焊接研究所(PWI)于20世纪60年代提出 ,并引入到钛合金的焊接. 目前针对铝合金已提出的活性焊接方法主要有传统 A-TIG 焊 ,法国学者 Sire 等人<sup>[3]</sup> 开发的 FB-TIG 焊 ,以及兰州理工大学开发的 FZ-TIG 焊三种.

传统的 A-TIG 焊是焊前在待焊焊道表面涂覆一层很薄的表面活性剂,可使熔深显著增加.然而焊接铝合金时,很难同时保证熔深显著增加和焊缝表面成形良好<sup>[4]</sup>. FB-TIG 焊是将活性剂涂敷于待焊焊道两侧,中间留出一定间隙,然后进行常规 TIG 焊,使得电弧弧根收缩,熔深增加. FB-TIG 焊虽能保证焊缝表面成形良好,但熔深增加程度不够理想. FZ-TIG 焊是焊前在待焊焊道表面中心区域涂敷低熔、沸点和低电阻率的活性剂,在两侧区域分别涂敷高熔、沸点和高电阻率的活性剂,然后进行常规 TIG焊,河同时保证焊接熔深增加和焊缝表面成形良好,可同时保证焊接熔深增加和焊缝表面成形良好。可同时保证焊接熔深增加和焊缝表面成形良好。可同时保证焊接熔深增加和焊缝表面成形良好。可同时保证焊接熔深增加和焊缝表面成形良好。可同时保证焊接熔深增加和焊缝表面成形良好。则是同上述两种方法一样,该方法仍然需要人工或机械涂敷活性剂,涂敷厚度不易精确控制,导致焊接生产效率降低,焊缝成形不易控制.

针对铝合金,提出一种新型活性 TIG 焊接方法—气体输送活性钨极氩弧焊,即 GTFA-TIG 焊(gas

transfer flux activating TIG welding). 该方法改变了活性元素的引入方式,由自动送粉装置通过保护气体将活性剂输送到焊接电弧—熔池系统中,从而收缩焊接电弧,改变熔池金属流态,增加焊缝熔深.

### 1 试验方法

试验选用的铝合金为 3A21,试件尺寸为 150 mm  $\times 80$  mm  $\times 8$  mm. 分别采用交流 TIG 焊、GTFA-TIG 焊进行表面熔焊 ,工艺参数分别如表 1 和表 2 所示. GTFA-TIG 焊的试验装置如图 1 所示 ,送粉器 为螺旋式送粉器 ,通过电机传动能够自动输送定量活性 剂. 活性 剂 采 用  $MnCl_2$  ,  $AIF_3$  ,  $MgF_2$  ,  $SiO_2$  ,  $AI_2O_3$  , $V_2O_5$  和  $TiO_2$  等 7 种常见的卤化物和氧化物以及单质碲 ,活性剂粒度为 100 目  $\sim 200$  目.

GTFA-TIG 焊前,首先加热活性剂,使活性剂吸附的水分以及本身的结晶水完全脱去,研磨并筛分. 试样先用丙酮擦拭 除去表面油污,然后再将试样用  $5\% \sim 10\%$  NaOH 溶液清洗  $3\sim 7$  min,使用流动清水冲洗,再用 30% HNO<sub>3</sub>溶液中和  $1\sim 3$  min,同样用流动清水冲洗,最后使用吹风机吹干试样. 焊接开始的同时输送活性剂粉末,熄弧后停止送粉.

采用 8 种单组元活性剂进行交流 GTFA-TIG 焊,分析不同活性剂对焊缝成形、焊缝拉伸力学性能和焊缝缺陷的影响,与传统交流 TIG 焊对比研究该新型活性焊接方法的可行性.

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#### 表 1 交流 TIG 焊工艺参数

Table 1 Specification of AC TIG welding

 焊接电流	焊接电流 焊接速度		弧长      氩气流量		———— 钨极伸出长度	
I/A	$v/(\text{ mm} \cdot \text{min}^{-1})$	$L/\mathrm{mm}$	$q/(L \cdot min^{-1})$	$d/\mathrm{mm}$	$L_1$ / mm	
130	100	3	10	3.2	3	

#### 表 2 交流 GTFA-TIG 焊工艺参数

Table 2 Specification of AC GTFA-TIG welding

 焊接电流	焊接速度	弧长	氩气流量	电机转速	钨极直径	钨极伸出长度
I/A	$v/(\text{ mm} \cdot \text{min}^{-1})$	$L/\mathrm{mm}$	$q/(L^{\bullet} min^{-1})$	$n/(r \cdot min^{-1})$	$d/\mathrm{mm}$	$L_1$ /mm
130	100	3	10	30	3.2	3

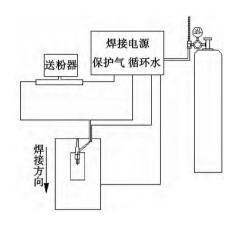


图 1 GTFA-TIG 焊试验装置示意图

Fig. 1 Schematic of GTFA-TIG welding

# 2 试验结果

#### 2.1 焊缝成形结果

交流 TIG 和 GTFA-TIG 焊的焊缝表面和横截面形貌如图 2 所示 焊缝尺寸如表 3 所示.

试验结果表明,除  $\mathrm{MgF_2}$  和  $\mathrm{TiO_2}$  以外, $\mathrm{MnCl_2}$  ,  $\mathrm{AlF_3}$   $\mathrm{SiO_2}$   $\mathrm{V_2O_5}$  ,  $\mathrm{Al_2O_3}$  这 5 种卤化物和氧化物增加熔深的效果都很好,都能使熔深达到传统  $\mathrm{TIG}$  焊熔深的  $2.5 \sim 3$  倍. 其中卤化物活性剂  $\mathrm{MnCl_2}$  和  $\mathrm{AlF_3}$  分别使熔深增加到传统  $\mathrm{TIG}$  焊的 3.0 倍和 2.8 倍,氧化物活性剂  $\mathrm{SiO_2}$  ,  $\mathrm{V_2O_5}$  和  $\mathrm{Al_2O_3}$  分别使熔深增加到传统  $\mathrm{TIG}$  焊熔深的 3.1 倍、2.8 倍和 2.5 倍,而单质碲增加熔深的效果则相对较差,只使熔深增加到传统  $\mathrm{TIG}$  焊熔深的 1.9 倍,从焊缝表面成形来看,采用  $\mathrm{MnCl_2}$ 、碲活性剂的  $\mathrm{GTFA-TIG}$  焊焊缝成形最好  $\mathrm{SiO_2}$  ,  $\mathrm{AlF_3}$  ,  $\mathrm{Al_2O_3}$  活性剂的其次,而  $\mathrm{V_2O_5}$  ,  $\mathrm{MgF_2}$  和  $\mathrm{TiO_2}$  的则较差.

#### 2.2 焊缝拉伸力学性能分析

按国家标准 GB/T228—2002 ,对母材以及采用  $MnCl_2$  , $AlF_3$  和  $V_2O_5$  活性剂的 GTFA—TIG 焊缝进行 了拉伸试验 ,试样尺寸如图3所示 ,拉伸速度为0.5

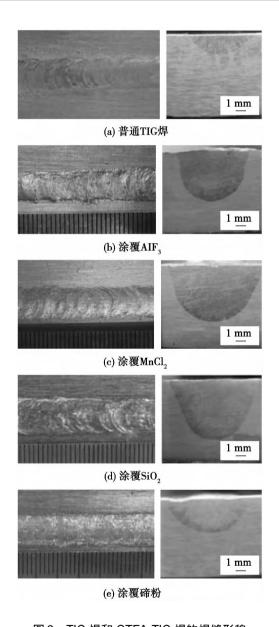


图 2 TIG 焊和 GTFA-TIG 焊的焊缝形貌 Fig. 2 Weld appearances of TIG and GTFA-TIG welding

mm/min. 采用碲的 GTFA-TIG 焊缝由于存在气孔,不进行拉伸试验. 拉伸结果如表 4 所示.

试验结果表明 采用卤化物 MnCl<sub>2</sub> 和 AlF<sub>3</sub> 活性

表 3 交流 TIG 和 GTFA-TIG 焊缝尺寸 Table 3 Weld geometry of AC TIG and GTFA-TIG welding

方法	活性剂	熔深 D/mm	熔宽 W/mm	深宽比 n
普通 TIG 焊		1.81	6.02	0.30
	$AlF_3$	5.05	7.29	0.69
	$\mathrm{MnCl}_2$	5.48	8.91	0.62
	$\mathrm{MgF}_2$	2.79	7.38	0.38
CTEATIC #	$\mathrm{SiO}_2$	5.62	9.05	0.62
GTFA-TIG 焊	$V_2O_5$	5.14	9.77	0.53
	$Al_2O_3$	5.04	14. 25	0.35
	${ m TiO_2}$	2.93	7.57	0.39
	Те	3.35	8.35	0.40

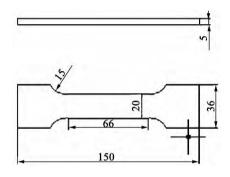


图 3 拉伸试样几何尺寸(mm) Fig. 3 Size of tensile sample

表 4 采用不同活性剂的 GTFA-TIG 焊缝抗伸试验结果
Table 4 Tensile test results of GTFA-TIG weld with different activating agent

	活性剂	断后伸长率	断面收缩率	抗拉强度
	油性剂	A( %)	Z(%)	$R_{\rm m}/{ m MPa}$
	MnCl <sub>2</sub>	0.135	0.615	98.57
GTFA-TIG 焊缝	AlF <sub>3</sub>	0.115	0.570	93.98
	$V_2O_5$	0.155	0.730	109.37
母材		0.110	0.650	116.01

剂的 GTFA-TIG 焊试样均断于焊缝区 ,最大抗拉强度较母材有一定的下降 ,分别达到母材的 85% 和 81% ,而采用氧化物  $V_2O_5$  活性剂的则均断于焊缝热影响区 ,最大抗拉强度达到母材的 94% .

#### 2.3 焊缝缺陷分析

选取了四种典型活性剂  $MnCl_2$  , $AlF_3$  , $V_2O_5$  和碲粉进行 GTFA-TIG 焊 ,每种活性剂焊两道焊缝 ,对所得的 8 条焊缝进行 X 射线探伤. 探伤参数为: 管电压 150 kV ,管电流 5 mA ,曝光时间 4 min. 探伤结果为: 除采用碲粉活性剂的焊缝评片结果为 II 级 ,存在气孔外 ,采用其它三种活性剂的焊缝评片结果均为 I 级. 说明采用卤化物  $MnCl_1$  , $AlF_3$  和氧化物  $V_2O_5$ 

活性剂的 GTFA-TIG 焊缝未发现气孔、夹渣、裂纹和条形缺陷 即采用 GTFA-TIG 焊接方法 ,可以获得质量良好的焊缝.

## 3 分析与讨论

在 GTFA-TIG 焊接中 ,MnCl, ,AIF, ,SiO, ,V,O, , Al<sub>2</sub>O<sub>3</sub> 都能使熔深达到传统 TIG 焊熔深的 2.5~3 倍以上 即使单质碲也能达到 1.9 倍. 而关于活性 TIG 焊熔深增加的作用机理,目前主要有电弧收缩 理论和 Marangoni 对流改变两种理论<sup>[6]</sup>. 以往的研 究表明[5] 对于铝合金 ,熔深增加的主要机理是电 弧收缩 而与 Marangoni 对流变化关系不大,这是由 于铝合金表面张力较低 热导率较高 以致材料表面 张力温度系数和温度梯度都较低,从而导致 Marangoni 对流较弱,因此,对增加铝合金焊缝熔深的 作用不大. 普通交流 TIG 焊和 GTFA-TIG 焊电弧形 貌如图 4 所示,试验过程中发现,相同参数下 GTFA-TIG 焊电弧电压比普通 TIG 焊的高 3~6 V 左右,由 此可证明 GTFA-TIG 焊电弧的确发生了强烈收缩. 对于铝合金 GTFA-TIG 焊 ,电弧收缩应该是熔深增 加的主要机理.



(a) 普通交流TIG焊

(b) GTFA-TIG焊

图 4 电弧形貌 Fig. 4 Appearances of welding arc

表 5 元素的电子亲和能(kJ/mol)
Table 5 Electron affinity energy of element

Ar	O	Cl	F	Te
0	141.0	348.8	327.9	190.4

O Cl F 和 Te 元素有较高的电子亲和能 ,容易捕捉电弧外围电子 ,使电弧收缩 . 另外卤化物和氧化物为多原子分子 在电弧作用下发生热解离 ,而热解离为吸热反应 根据最小电压原理 ,电弧收缩 . 几种活

性剂的标准摩尔生成焓如表 6 所示<sup>[7]</sup>. 电弧在上述两种作用下发生强烈收缩,使得电压增加,热输入增大,同时使电弧和熔池内的电磁力增大,电弧热量更有效传输到熔池底部,焊缝熔深显著增加.

表 6 活性剂标准摩尔生成焓
Table 6 Standard molar formation enthalpy of flux

活性剂	$\mathbf{MnCl}_2$	$\mathrm{AlF}_3$	$\mathrm{MgF}_2$	$\mathrm{SiO}_2$	$\mathrm{V}_2\mathrm{O}_5$	$\mathrm{Al}_2\mathrm{O}_3$	${ m TiO_2}$
$\Delta H_{\rm f~298}^{\theta}$ /( kJ·mol -1)	-481.997	-1 510.424	-1 123.404	-908.346	-1 557.703	-1 675.700	- 944. 747

# 4 结 论

- (1) 针对铝合金材料,提出了一种新型活性 TIG 焊接方法—气体输送活性钨极氩弧焊,该方法 改变了活性元素的引入方式,通过自动送粉装置将 适量的活性剂输送到焊接保护气体中,由保护气体 将活性剂引入电弧—熔池系统进行施焊.可同时保 证焊缝熔深成倍增加和焊缝表面成形良好,并且省 却了涂覆活性剂工序,实现了焊接过程自动化.
- (2) 除  $MgF_2$  和  $TiO_2$  以外 , $MnCl_2$  , $AlF_3$  , $SiO_2$  ,  $V_2O_5$  , $Al_2O_3$  这 5 种卤化物和氧化物增加熔深的效果都很好 ,都能使熔深达到传统 TIG 焊的 2. 5 ~ 3 倍 ,而单质碲增加熔深的效果则相对较差. 采用  $MnCl_2$  和碲粉的 GTFA–TIG 焊焊缝成形最好 , $SiO_2$  ,  $AlF_3$  和  $Al_2O_3$  活性剂的其次 ,而  $V_2O_5$  , $MgF_2$  和  $TiO_2$  的则较差.
- (3) 采用卤化物  $MnCl_2$  和  $AlF_3$  活性剂的 GT-FA-TIG 焊缝最大抗拉强度分别达到母材的 85% 和 81% 而采用氧化物  $V_2O_5$  活性剂的焊缝最大抗拉强度最高 达到母材的 94%.
- (4) 采用卤化物  $MnCl_2$  , $AlF_3$  和氧化物  $V_2O_5$  活性剂的 GTFA-TIG 焊缝经 X 射线探伤未发现气孔、夹渣、裂纹和条形缺陷 ,而采用碲粉的焊缝则发现气孔.

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The effect of nano-Sb addition on the growth kinetics of intermetallic compound (IMC) in the joints with Sn-3. 0 Ag- 0.5 Cu- x Sb ( x = 0 , 0.2 , 1.0 , and 2.0% ) solder in reflow process was investigated. Scanning electron microscope ( SEM) was used to observe the microstructure evolution of soldered joints, energy dispersive X-ray (EDX) and x-ray diffractometer (XRD) were used to identify the IMC phases. The results show that some nano-Sb particles were dissolved in Sn-rich phase, some participated in the formation of Ag<sub>3</sub>Sb, and the rest dissolved in Cu<sub>6</sub>Sn<sub>5</sub> IMC layer. The thickness of IMC layer decreased when the nano-Sb was added. Under the isochronal conditions , the thickness of IMC layer in all soldered joints was minimum when the amount of nano-Sb increased to about 1.0%. The growth exponent n and diffusion coefficient D for IMC layer were determined by curve-fitting. The results reveal that the growth exponent n and diffusion coefficient D decreased with the increase of nano-Sb content. When the amount of nano-Sb increased to about 1.0% , the growth exponent n and diffusion coefficient D were 0.326 and 10.31  $\times$  10<sup>-10</sup> cm<sup>2</sup>/s, respectively. Based on the phase diagram analysis and adsorption theory, Sb had better affinity to Sn, and it could reduce the activity of Sn by forming SnSb compound, resulting in decreased driving force and surface energy for formation of Cu-Sn IMC. Therefore, the growth rate of Cu<sub>6</sub>Sn<sub>5</sub> grains was suppressed and the growth of IMC layer was retarded.

**Key words**: reflow soldering; nano-Sb dopant; lead free solder; intermetallic compound

Gas transfer flux activating TIG welding process for aluminum alloy HUANG Yong<sup>1,2</sup>, LI Tao<sup>2</sup>, WANG Yanlei<sup>2</sup> (1. State Key Laboratory of Gansu Advanced Non-ferrous Metal Materials, Lanzhou University of Technology, Lanzhou 730050, China; 2. Key Laboratory of Non-ferrous Metal Alloys, The Ministry of Education, Lanzhou University of Technology, Lanzhou 730050, China). pp 101 – 104

A new activating TIG welding process, gas transfer flux activating TIG (GTFA-TIG) welding, is proposed for welding aluminum alloy by changing the introduction method of active elements. In this process, a flux is transferred to shielding gas through an automatic powder feeder, and the shielding gas carries the active elements into the welding arc and weld pool. In this way, the welding arc is contracted, and the flow pattern in the weld pool is changed, finally, the weld depth increases. Without coating the flux, the automation of welding process is realized. Compared to the conventional AC TIG welding , the effects of fluxes on the weld shape , tensile property and defects of GTFA-TIG welding with eight kinds of single-component fluxes were investigated. It proves that most of halide and oxide fluxes can improve the weld depth up to 2.5-3.0 times of that of traditional TIG welding, while the effect of Te flux is less effective. The tensile strength of the joint with V<sub>2</sub>O<sub>5</sub> flux is close to that of base metal , but it is lower than that of base metal when using MnCl<sub>2</sub> and AlF<sub>3</sub> fluxes. The X-ray detection of defects shows that the welds with V2O5, MnCl2 and AlF3 fluxes are assessed as grade I, while that with Te flux is grade III.

**Key words**: GTFA-TIG welding; aluminum alloy; arc contraction; activating flux

Microstructure and mechanical properties of welded joint of nanostructured bainite steel with regeneration  $\rm FANG~Kun^1$ , HUANG  $\rm Nan^2$ , YANG Jianguo  $^{1.3}$ , SONG Kuijing  $^{1}$ , FANG Hongyuan  $^{1}$ ( 1. State Key Laboratory of Advanced Welding and Joining , Harbin Institute of Technology , Harbin 150001 , China; 2. Armor Force Technique Institute of PLA Teaching and Training Department , Changchun 130117 , China; 3. Institute of Process Equipment and Control Engineering , Zhejiang University of Technology , Hangzhou 310032 , China) . pp 105 – 108

Abstract: Due to the poor weldability of nanostructured bainite steel , a method called welding with regeneration was proposed to obtain a welded joint with the same microstructure as the base metal , so the welded joint could possess the excellent mechanical properties. Regeneration treatment was used after tungsten inert gas (TIG) welding of nanostructured bainite steel. The microstructures of the welded joint were characterized by X-ray diffraction , scanning electron and transmission electron microscopes. The tensile and microhardness tests were carried out to evaluate the mechanical properties. The results show that nanostructured bainite was obtained in the fusion zone and quenched zone with 300 nm and 70 nm thick bainite plates , respectively. The tensile strength of the welded joint was 1 500 MPa , and the microhardness in the fusion zone was 7 000 MPa , which were almost equivalent to those of the base metal.

**Key words**: nanostructured bainite steel; regeneration; welding; nano-scale plate; strength

Quality assessment for resistance spot welding based on Bayesian image recognition technology ZHANG Hongjie<sup>1,2</sup>, ZHANG Jianye<sup>1,2</sup>, SUI Xiuwu<sup>1,2</sup> (1. School of Mechanical Engineering, Tianjin Polytechnic University, Tianjin 300387, China; 2. Tianjin Key Laboratory of Modern Mechatronics Equipment Technology, Tianjin Polytechnic University, Tianjin 300387, China). pp 109 – 112

Abstract: A method for converting electrode displacement signal to binary image in resistance spot welding process is proposed. Based on image characteristic analysis, fifteen image features are extracted from the binary image of electrode displacement waveform. The principal component analysis is used to remove the cross correlation among image features,, and a series of weld specimens with different welding quality are selected to develop a quality classifier. The test results based on Bayesian image recognition technology of minimum risk show that it is feasible and reliable to utilize the binary image of electrode displacement signal to evaluate the weld quality and the image conserves the information of the weld quality. The algorithm for image feature extraction is simple, efficient and easy to use. At the mean time, the Bayesian image recognition technique with small samples can realize the welding quality assessment rapidly and accurately, and the method has a broad application prospect.

**Key words**: resistance spot welding; quality assessment; Bayesian statistic analysis; image recognition