



Ionic liquid-derived Co_3O_4 /carbon nano-onions composite and its enhanced performance as anode for lithium-ion batteries

Yanshuang Meng^{1,3}, Gongrui Wang¹, Mingjun Xiao¹, Chaoyu Duan¹, Chen Wang¹, Fuliang Zhu^{1,3,*}, and Yue Zhang^{2,*}

¹School of Materials Science and Engineering, Lanzhou University of Technology, Lanzhou 730050, China

²Department of Mechanical and Industrial Engineering, Texas A&M University-Kingsville, Kingsville, TX 78363, USA

³State Key Laboratory of Advanced Processing and Recycling of Non-ferrous Metals, Lanzhou 730050, China

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ABSTRACT

In this work, a novel composite of Co_3O_4 nanoparticle and carbon nano-onions (CNOs) is synthesized by using ionic liquid as carbon and nitrogen source through a facile carbothermic reduction followed by low-temperature oxidation method. The SEM and HRTEM images reveal that the Co_3O_4 particles are homogeneously embedded in the CNOs. Due to the unique nano-structure, the electrolyte contacts well with the active materials, leading to a better transfer of lithium ions. Moreover, the unique nano-structure not only buffers the volume changes but also facilitates the shuttling of electrons during the cycling process. As a result, the electrode made up of Co_3O_4 /CNOs composite delivers favorable cycling performance (676 mAh g^{-1} after 200 cycles) and rate capability (557 mAh g^{-1} at the current of 1 C), showing a promising prospect for lithium-ion batteries as anode materials.

Introduction

Lithium-ion batteries (LIBs) have been identified as one of the most important energy storage devices as their applications have been successfully expanded from small-scale portable electronics to large-scale electronic vehicles [1]. Previous researches indicated that the electrode material, especially the anode material, plays a critical role in determining the performance of LIBs [2, 3]. Graphite is a commonly used anode material in commercial LIBs; however, a

theoretical capacity of only 372 mAh g^{-1} significantly limited its LIB performance [4]. Conversion reaction-based transition metal oxides, including SnO_2 [5], CuO [6], Fe_3O_4 [7], and Co_3O_4 [8], are attractive candidates for LIB anodes. Among these metal oxides, Co_3O_4 has received extensive attention of researches, because of its many beneficial properties such as excellent reversible redox behavior, good corrosion stability, its simple preparation method, and promising ratio of surface atoms and diverse morphologies [9–11]. More importantly, it shows

Address correspondence to E-mail: chzfl@126.com; yue.zhang@tamuk.edu

huge development potential due to its high theoretical capacity (890 mAh g^{-1}) when compared with conventional graphite electrodes [12, 13]. However, the wide application of Co_3O_4 electrodes is hindered by its low inherent conductivity and large volume variation during lithiation/delithiation, which can lead to pulverization, electrochemical sintering, and eventually a capacity fading [14, 15].

To overcome these shortcomings, numerous methods have been investigated, including nano-crystallization [16], combination with other oxide materials [17, 18], and incorporation with carbon materials [19–21]. Among these methods, incorporating carbon materials with Co_3O_4 becomes the focus of researches recently as this method exhibits the potential to achieve the best electrochemical and dynamic performance. The integrated carbon materials, such as reduced graphene oxide [22, 23], carbon nanotubes [20], carbon nanofiber [24, 25], hollow carbon spheres [26], porous carbon [27], and onion-like carbon [28–30], can not only buffer the volume changes but also facilitate the shuttling of electrons, leading to an enhanced electrical conductivity during cycling process.

Ionic liquids (ILs) have been studied to a lesser extend as carbon source but are gaining increased interests recently for applications in LIBs due to its negligible vapor pressure and designable structure [31–35]. The negligible vapor of ILs moderates the evaporation of ILs during the carbonizations process, which makes the processing and shaping process simple to implement. The designable structure allows a controllable ratio of cation and anion components in ILs, which can tune the doping contents of heteroatoms in the carbon materials.

In this work, we have produced $\text{Co}_3\text{O}_4/\text{CNOs}$ composites by using 1-butyl-3-methylimidazolium dicyanamide ($\text{BMIm})\text{N}(\text{CN})_2$ as carbon and nitrogen

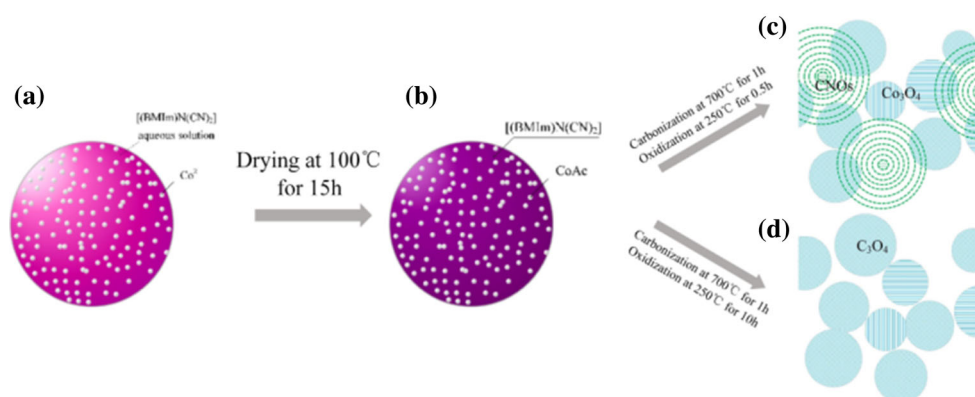
source. The composite was first characterized by a variety of techniques including powder X-ray diffraction (XRD), Raman spectroscopy, X-ray photoelectron spectroscopy (XPS), and electron microscopy to determine the quality of the electrode materials. The material was assembled into coin-type half cells to measure the electrochemical properties. Because of the onion-like structure of the carbon component, the interfaces of carbon and Co_3O_4 particles are well developed, leading to a symmetrical acceleration of electron shuttling and subsequently improved electrochemical performance.

Experimental section

Material synthesis

The $\text{Co}_3\text{O}_4/\text{CNOs}$ composite is synthesized through direct pyrolysis of ionic liquid precursor and cobalt(II) acetate tetrahydrate ($\text{C}_4\text{H}_6\text{CoO}_4 \cdot 4\text{H}_2\text{O}$, Aladdin AR 99.5%), followed by low-temperature oxidation. In a specific preparation method, 0.517 g $\text{C}_4\text{H}_6\text{CoO}_4 \cdot 4\text{H}_2\text{O}$ is dissolved in 2 ml ultrapure water, followed by adding 1 g of $(\text{BMIm})\text{N}(\text{CN})_2$ (purchased from Lanzhou Institute of Chemical Physics, Chinese Academy of Sciences). After that, the mixture is ground in the agate mortar for 30 min at room temperature (Fig. 1a). The material is transferred into a ceramic crucible and dried at 100°C for 15 h in the air blowing thermostatic oven (Fig. 1b). The as-obtained material is annealed at 350°C for 30 min and then at 700°C for 1 h in the tube furnace with the protection of an argon atmosphere. The material is oxidized by heating treatment at 250°C for 0.5 h for $\text{Co}_3\text{O}_4/\text{CNOs}$ composite and 10 h for bare Co_3O_4 in the ambient atmosphere, respectively (Fig. 1c).

Figure 1 Schematic of the synthesis process for $\text{Co}_3\text{O}_4/\text{CNOs}$ composites and bare Co_3O_4 .



Material characterization

The phase composition of the as-synthesized material is characterized by powder XRD (D/max-2400, Cu K α). Further information of the materials is investigated using Raman on a LabRAM HR800 equipment and the XPS on a Kratos XSAM 800 spectrometer. The morphology and microstructure are determined by scanning electron microscope (SEM) on a Quanta450FEG equipment and transmission electron microscopy (TEM) and high-resolution TEM (HRTEM) on a JEM-1200EX equipment. The carbon and nitrogen element contents are characterized by Elementar Analysensysteme GmbH VarioEL cube. The Brunauer–Emmett–Teller nitrogen adsorption-desorption measurement of the as-synthesized material is taken on a Micromeritics ASAP 2020 equipment.

Electrochemical measurement

The electrochemical performance of the as-obtained material is tested by using CR2032-type coin cells fabricated in an argon-filled glove box. The coating slurry is manufactured by active materials, acetylene black, and polyvinylidene fluoride (PVDF) with a mass ratio of 7:2:1. The working electrode is fabricated by painting the slurry onto copper foils and then dried at 100 °C for 10 h in a vacuum oven.

The cyclic voltammetry (CV) measurement is employed to characterize the reaction potential and reaction type on a CHI660E electrochemical workstation at a scan rate of 0.1 mV s⁻¹ in a voltage range of 0–3 V (vs. Li/Li⁺). The capacity and cycling performances are measured by galvanostatic discharge/charge measurement with a specific current of 89 mA g⁻¹ in the voltage range of 0.01–3 V (vs. Li/Li⁺) on a battery test system (LAND CT2001A, China). The rate properties are carried out at various specific current densities 0.1, 0.2, 0.3, 0.5, 1, and 0.1 C.

Results and discussion

The XRD patterns of the Co₃O₄/CNOs composite and bare Co₃O₄ are shown in Fig. 2a. Both samples exhibit peaks at 18.8°, 31.1°, 36.9°, 38.2°, 44.8°, 55.4°, 59.4°, 65.2°, and 77.5°, which corresponds to the face-centered cubic phase of spinel-structured Co₃O₄ (JCPDS No. 42-1467) [36, 37]. In the spectrum of Co₃O₄/CNOs composite, the compressed diffraction peak (as

shown in Fig. 2b) at 23.8° is designated as the (002) reflection of graphite (JCPDS No. 41-1487) [38, 39]. When compared with the Co₃O₄/CNOs composite, the bare Co₃O₄ sample has sharper peaks, indicating that extended heating can facilitate to achieve a high phase purity and crystal degree of Co₃O₄.

The Raman spectrum of Co₃O₄/CNOs composite further confirms the formation of carbon (Fig. 2c). The broad Raman scattering peaks located at 1358 and 1587 cm⁻¹ can be assigned to the D and G band, arising from the sp³ and sp² hybridization vibrations of carbon materials, respectively [13]. No D or G band is observed in the spectrum of bare Co₃O₄, indicating that the carbon material is fully removed after heating in air for 10 h. The distinct peaks at 196, 481, 524, 619, and 686 cm⁻¹ are attributed to the B_{1g}, E_g, F_{2g}¹, F_{2g}², and A_{1g} vibration mode of Co₃O₄, respectively [12, 13, 40], certifying the existence of Co₃O₄ in both samples.

The XPS analysis, shown in Fig. 3, shows the near-surface chemical information of the Co₃O₄/CNOs composite. The convolution Co 2p spectrum is exhibited in Fig. 3a. The peaks centered at 780.6 and 795.9 eV are ascribed to Co 2p_{3/2} and Co 2p_{1/2} spin-orbital peaks of Co₃O₄, respectively [41]. This indicates that the Co species in the composites exist in the form of Co₃O₄, which is in agreement with the XRD and Raman results. The strong shake-up peaks located at 795.9 and 804.2 eV further confirm the presence of Co₃O₄ [42]. The spectrum of O 1s can be split into two peaks at 530.1 and 531.3 eV, corresponding to the Co–O and C–O bonds, respectively (Fig. 3b) [43]. As displayed in Fig. 3c, the high-resolution XPS spectrum of C 1s is deconvoluted into three peaks at 284.7, 285.6, and 286.5 eV, respectively. The peak centered at 284.7 eV is assigned to the sp² hybridized carbon, corresponding to C–C bonds or C=C bonds [44]. The peak located at 285.6 eV is ascribed to C=N bonds or C–O bonds [15, 45], which may arise from the nitrogenous carbon precursors and C–O–Co bonds that combine carbon materials with Co₃O₄ [46]. The peak at 286.5 eV is designated to C–N or C=O bonds [45, 47, 48]. Figure 3d exhibits the N 1s XPS spectrum, which indicates the presence of pyridinic N (398.7 eV), pyrrolic N (399.6 eV), and graphitic N (400.1 eV) [42]. Previous researches have reported that N-doped carbon can enhance conductivity, electrochemical activity, and stability [15, 49]. Particularly, the pyridinic and graphitic nitrogen atoms can facilitate the absorption of lithium ions, which

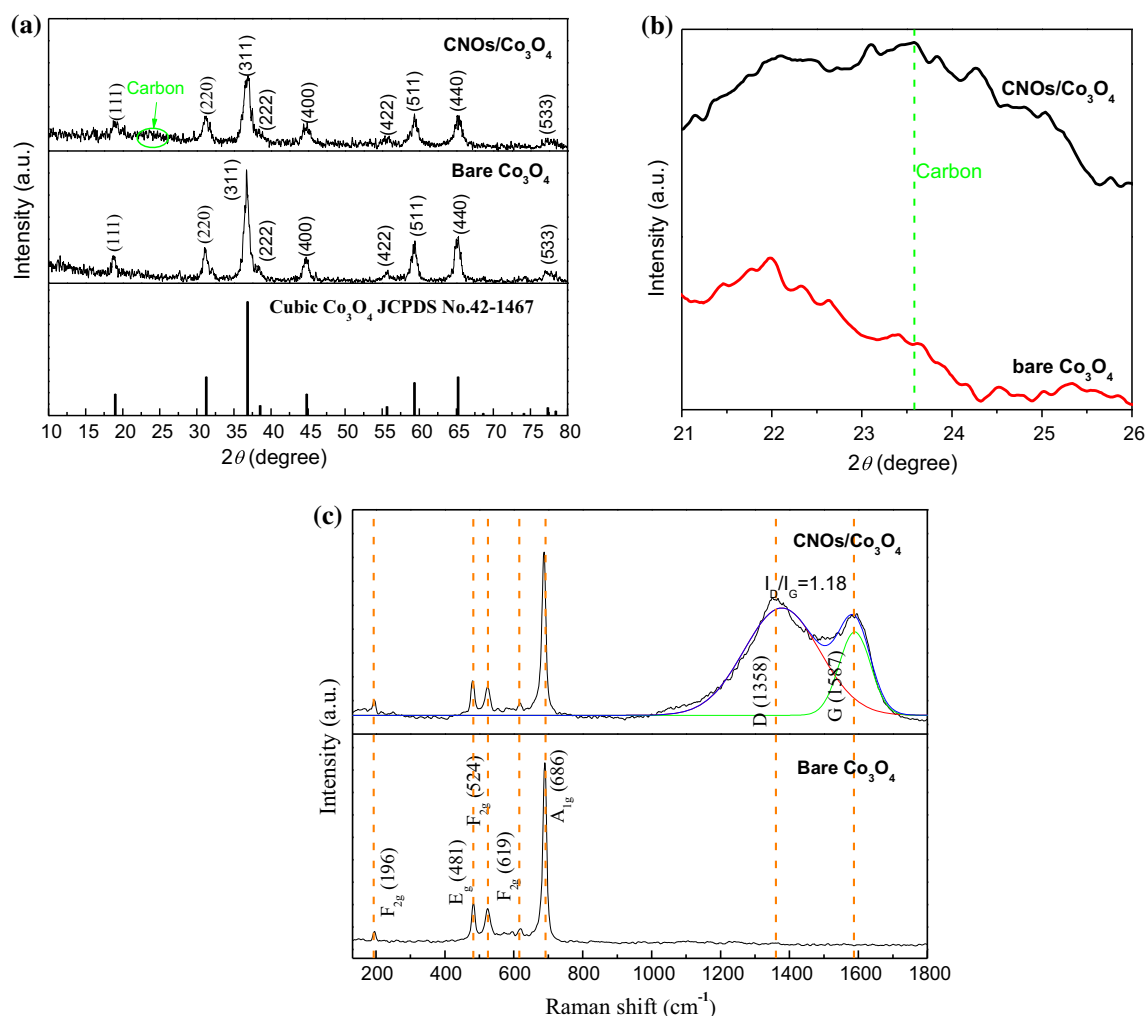


Figure 2 XRD patterns (a, b) and Raman spectra (c) of Co₃O₄/CNOs composite and bare Co₃O₄.

contributes to the capacity of the as-synthesized Co₃O₄/CNOs composite [50].

The electron microscopy is applied to observe the morphology of the Co₃O₄/CNOs composites (Fig. 4). As displayed in Fig. 4a, the SEM image reveals a nano-structure with Co₃O₄ embedding in CNOs, which is accompanied with trivial agglomeration of the particles with uniform distributed grain size. Due to this unique nano-structure, the electrolyte contacts well with the active materials, leading to a better transfer of lithium ions. Figure 4b and its inset present the TEM image and selected area electron diffraction (SEAD) patterns of as-prepared Co₃O₄/CNOs composites. The TEM image provides further information of the composites, showing that the particles observed in SEM image are constituted of smaller particles. The SEAD patterns exhibit the

typical diffraction rings of Co₃O₄ which is denoted by blue line. The diffraction ring of carbon, denoted by green line, is faint and broadened, which indicates the content of CNOs is low. A close view of region 1 and region 2 in Fig. 4b is shown in Fig. 4c, d. From the test of Fourier transform algorithm, it is found that the particles marked with blue dashed circle in Fig. 4c, d are Co₃O₄ particles. The particles in Fig. 4c display an interplanar spacing of 0.244 nm, corresponding to the (311) plane of spinel-structured Co₃O₄ (JCPDS No. 42-1467). The materials in the green dashed circle are carbon that exists in the form of CNOs. The formation of CNOs is because of the catalysis effects of Co₃O₄ particles and metallic cobalt that is formed in the pilot process [51, 52]. Due to the surface energy minimization effect, the CNOs are in the form of sphere. The Co₃O₄ particles are

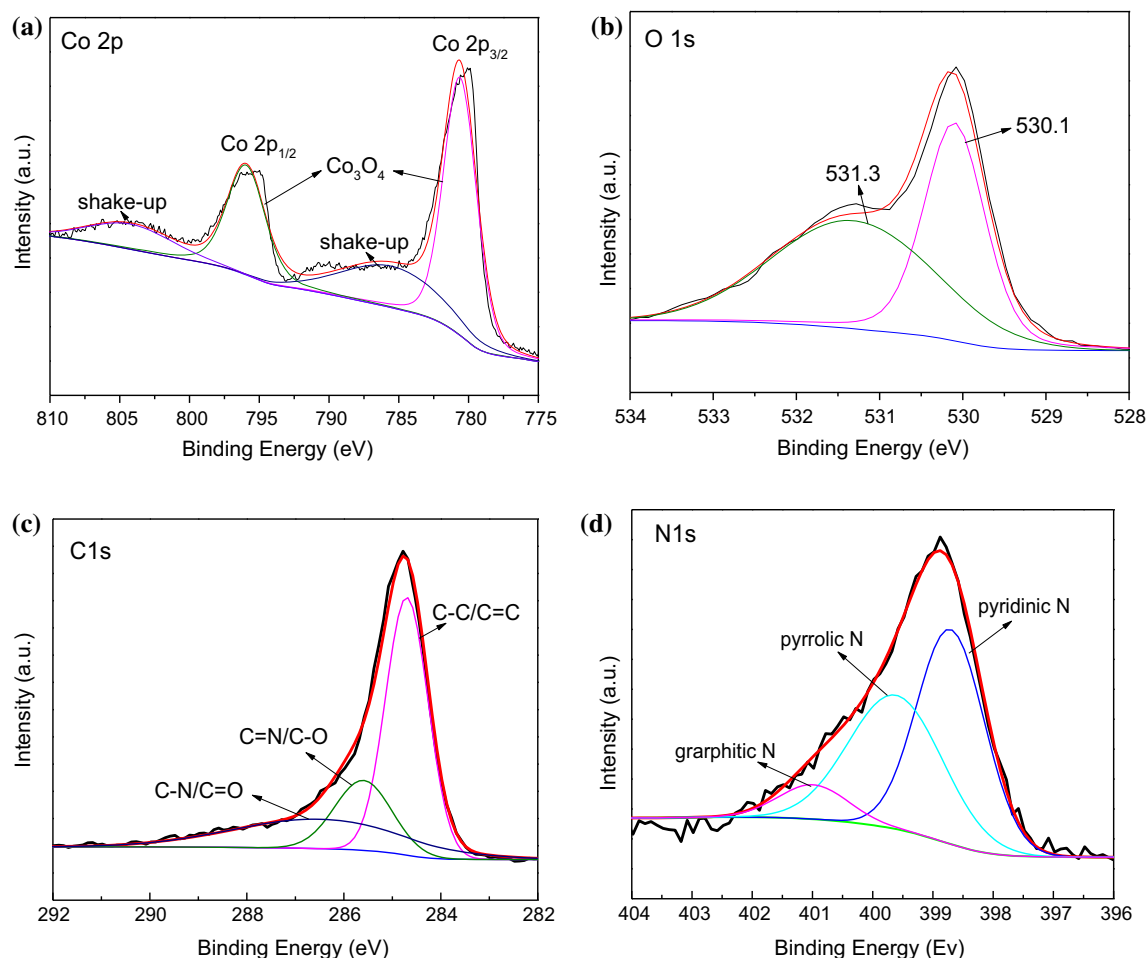


Figure 3 High-resolution XPS spectra of $\text{Co}_3\text{O}_4/\text{CNOs}$ composite: **a** Co2p, **b** C1s, **c** N1s, and **d** O1s.

embedded in the CNOs and partially expose to the external environment due to the oxidation process above 300 °C which evaporates carbon materials.

As shown in Fig. 5, N_2 adsorption/desorption experiments are performed to examine the pore size distribution and surface areas of the $\text{Co}_3\text{O}_4/\text{CNOs}$ composite. The isotherms of $\text{Co}_3\text{O}_4/\text{CNOs}$ composite are typical type curves according to IUPAC classification. The pore size distribution shows that the $\text{Co}_3\text{O}_4/\text{CNOs}$ composite possessed a mesoporous structure with an average pore size of 3.2 nm. The BET surface area of $\text{Co}_3\text{O}_4/\text{CNOs}$ composite is confirmed to be $79.79 \text{ m}^2 \text{ g}^{-1}$ which is in line with the observation in SEM image.

To investigate the electrochemical performance of the $\text{Co}_3\text{O}_4/\text{CNOs}$ composite, $\text{Co}_3\text{O}_4/\text{CNOs}$ -Li half battery is assembled, with the $\text{Co}_3\text{O}_4/\text{CNOs}$ composite and metallic lithium as positive and negative electrodes, respectively. Figure 6a shows the first

three CV curves of $\text{Co}_3\text{O}_4/\text{CNOs}$ composite at a scan rate of 0.1 mV s^{-1} , over the voltage range of 0–3 V. A distinct and irreversible reductive peak at 0.89 V (A) is observed during the first cathodic scan. This peak is arisen by two reactions: (1) the electrochemical reduction reaction of Co_3O_4 and the formation of amorphous Li_2O ($\text{Co}_3\text{O}_4 + 8\text{Li}^+ + 8\text{e}^- \leftrightarrow 4\text{Li}_2\text{O} + 3\text{Co}^0$) and (2) the reductive decomposition reaction of LiPF_6 and organic solvent (EC and DEC) in which the solid electrolyte interphase (SEI) film is formed on the surface of the $\text{Co}_3\text{O}_4/\text{CNOs}$. The cathodic peak around below 0.47 V (C) stems from the intercalation of Li^+ ions into CNOs of the $\text{Co}_3\text{O}_4/\text{CNOs}$ composite [53], indicating the electrochemical activity of CNOs. In the first anodic scan, the unambiguous peaks at 1.32 V (A') and 2.07 V (A'') are distinguished and designated to the conversion reaction of $\text{Co}^0 + x\text{Li}_2\text{O} \leftrightarrow \text{CoO}_x + 2x\text{Li}^+ + 2\text{e}^-$ (CoO_x refers to CoO or Co_3O_4), certifying that the

Figure 4 SEM (a), TEM (b), HRTEM images (c, d), and SAED patterns (inset of b) of $\text{Co}_3\text{O}_4/\text{CNOs}$ composite.

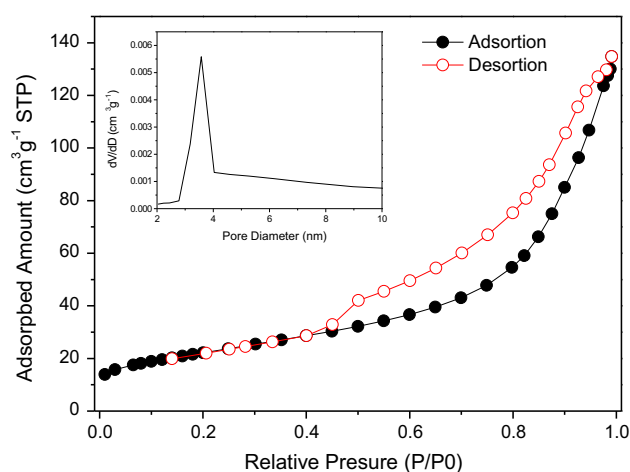
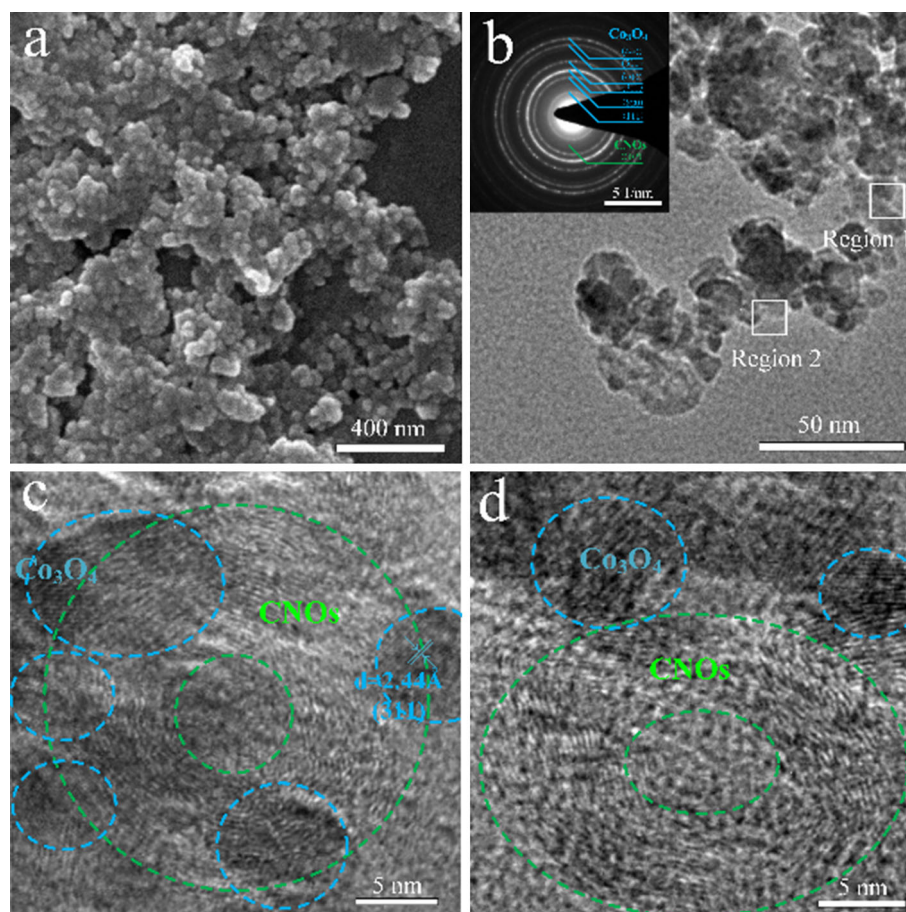


Figure 5 Nitrogen adsorption–desorption isotherms of the $\text{Co}_3\text{O}_4/\text{CNOs}$ composite and the pore size distribution curve (inset picture).

oxidization of Co^0 to Co_3O_4 is a multistep reaction [54, 55]. The reduction peak at 0.89 V (A) in the first cathodic scan shows a shift and is split into two

lower-intensity and broad peaks at 1.24 and 0.98 V, which are ascribed to the reduction reaction of Co_3O_4 and CoO [54, 56], respectively. The primary peaks of the third cycle are almost overlapped with the peaks of the second cycle, indicating a great reversibility and integrity of the electrode [57].

Figure 6b displays the discharge/charge profiles of the $\text{Co}_3\text{O}_4/\text{CNOs}$ composite at a specific current density of 89 mA g^{-1} . There is a long voltage plateau at 1.01 V in the initial discharge process, which is attributed to the formation of SEI layers on active materials and the electrochemical reduction reaction of Co_3O_4 [58]. The voltage declines slowly to the cut voltage of 0.01 V after the long voltage plateau and eventually results in a discharge capacity of 1468 mAh g^{-1} . After that, the charge process displays two identifiable and gradient voltage plateau at 1.32 and 2.06 V, which are ascribed to the oxidation reaction of metallic cobalt particles and the decomposition of Li_2O [54, 55, 59]. The following curves are almost overlapped, showing a favorable reversibility of the $\text{Co}_3\text{O}_4/\text{CNOs}$ composite.

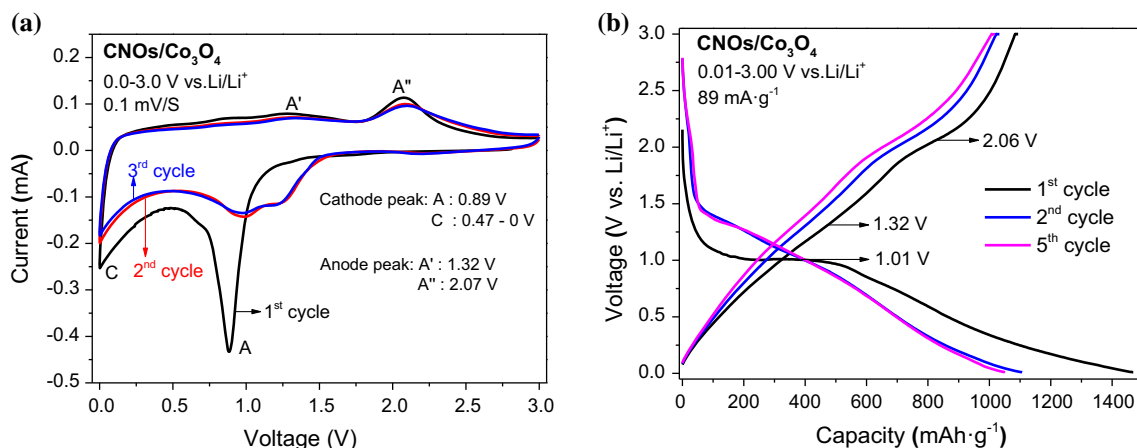


Figure 6 CV measurements (a) and the discharge/charge curves (b) of $\text{Co}_3\text{O}_4/\text{CNOs}$ composite.

Figure 7 Galvanostatic charge–discharge profiles of $\text{Co}_3\text{O}_4/\text{CNOs}$ composite and bare Co_3O_4 at 89 mA g^{-1} .

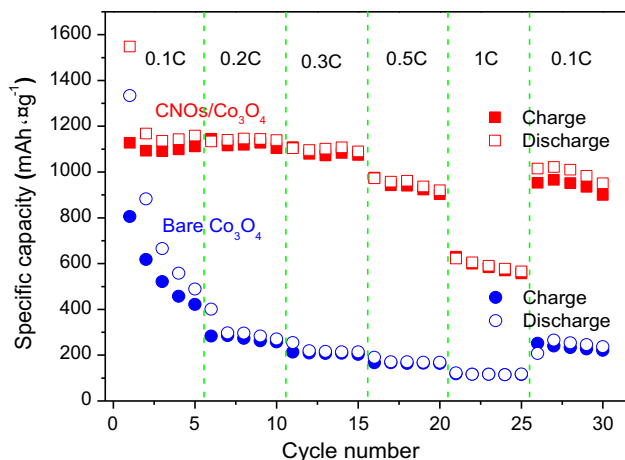
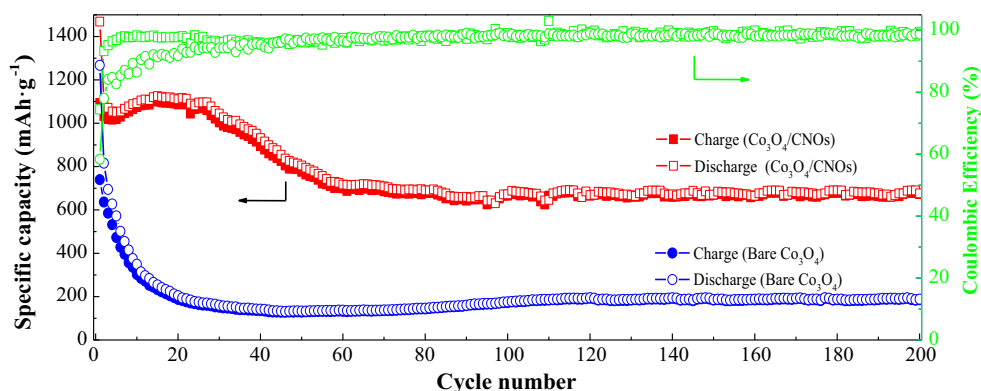


Figure 8 Rate performance of $\text{Co}_3\text{O}_4/\text{CNOs}$ composite and bare Co_3O_4 at various current densities.

As exhibited in Fig. 7, the cycling performance of the $\text{Co}_3\text{O}_4/\text{CNOs}$ composite and bare Co_3O_4 as anode materials is measured at a current density of 89 mA h g^{-1} over 0.1–3.00 V. The electrode of bare Co_3O_4 shows a reversible capacity of 172 mA h g^{-1} .

For the electrode of $\text{Co}_3\text{O}_4/\text{CNOs}$ composite, the first discharge and charge capacities are 1468 and 1093 mA h g^{-1} , respectively, with an initial coulombic efficiency of 74.47% . The initial discharge and charge capacities exceed the theoretical capacity of $\text{Co}_3\text{O}_4/\text{CNOs}$ composite, which is estimated as follows:

$$\begin{aligned} C_{\text{theoretical}} &= C_{\text{graphite}} \times \text{mass ratio of carbon materials} \\ &\quad + C_{\text{Co}_3\text{O}_4} \times \text{mass ratio of Co}_3\text{O}_4 \\ &= 372 \text{ mA h g}^{-1} \times 6.24\% + 890 \text{ mA h g}^{-1} \\ &\quad \times 93.76\% = 857.7 \text{ mA h g}^{-1} \end{aligned}$$

The content of CNOs in the composites is tested by elemental analysis. The extra capacities of $\text{Co}_3\text{O}_4/\text{CNOs}$ composite may be ascribed to the auxiliary lithium-ion storage sites, such as defects, grain boundaries of Li_2O , and the formation of metallic cobalt in the discharge process [36, 52, 60, 61]. Furthermore, the reversible formation/dissolution conversion reaction, following electrochemical catalytic conversion mechanism which refers to the effective catalysis of metallic cobalt and Co_3O_4 of SEI layers

during cycling, contributes to the extra discharge/charge capacities [60, 62]. The pyridinic and graphitic nitrogen atoms of $\text{Co}_3\text{O}_4/\text{CNOs}$ can contribute to the capacity by facilitating the absorption of lithium ions [50]. Moreover, the enlarged (002) interplanar distance can provide lithium-ion storage sites [63].

After the first cycle, the reversible discharge/charge capacities increase gradually to 1120 and 1100 mAh g^{-1} at 15th cycle, respectively. The increased capacities are due to the reversible formation of pseudocapacitive polymeric film and the incremental lithium-ion storage sites among amorphous CoO_x that are formed from the conversion of the crystal state during the cycling process [16, 64, 65]. The bare Co_3O_4 electrode displays a sharp decrease in capacity after the first cycle, while the capacities of the $\text{Co}_3\text{O}_4/\text{CNOs}$ electrode are stable and do not show any capacity decrease before the 30th cycle, which may be due to the carbon materials which serve as the protection layers of the Co_3O_4 particles. After the 30th cycle, the capacities of the $\text{Co}_3\text{O}_4/\text{CNOs}$ electrode gradually reduce and become steady after the 60th cycle. The capacities reduction may be due to the nano-structure exfoliation caused by the constant volume variations during the cycling process. Because of the special structure, the $\text{Co}_3\text{O}_4/\text{CNOs}$ electrode still performs a discharge and charge capacities of 676 and 672 mAh g^{-1} with a coulombic efficiency of 99% after 200 cycles.

As shown in Fig. 8, the rate performance of $\text{Co}_3\text{O}_4/\text{CNOs}$ composite and bare Co_3O_4 is measured at 0.1, 0.2, 0.3, 0.5, 1, and 0.1 C for five cycles at each current density. For the bare Co_3O_4 , the reversible capacities are 421, 257, 203, 163, 116, and 221 mAh g^{-1} at the current of 0.1, 0.2, 0.3, 0.5, 1, and 0.1 C, respectively. The $\text{Co}_3\text{O}_4/\text{CNOs}$ composite presents a reversible capacity of 1111, 1103, 1073, 903, 557, and 898 mAh g^{-1} at the current of 0.1, 0.2, 0.3, 0.5, 1, and 0.1 C, respectively. The rate performance of $\text{Co}_3\text{O}_4/\text{CNOs}$ composite is favorable compared with bare Co_3O_4 and cycling performance of $\text{Co}_3\text{O}_4/\text{CNOs}$ composite. When compared with the bare Co_3O_4 , the $\text{Co}_3\text{O}_4/\text{CNOs}$ composite exhibits better rate performance and cycling performance.

Conclusions

A novel $\text{Co}_3\text{O}_4/\text{CNOs}$ composite with homogenous nano-structure is synthesized via a facile method at low temperature by using $(\text{BMIm})\text{N}(\text{CN})_2$ as carbon

and nitrogen source. The $\text{Co}_3\text{O}_4/\text{CNOs}$ composite is consisted of Co_3O_4 nano-crystal and carbon nano-onions. The unique nano-structure can not only buffer the volume changes but also facilitate the shuttling of electron, which can promote the conductivity of the materials during cycling process. The electrode made up of $\text{Co}_3\text{O}_4/\text{CNOs}$ composite delivers favorable cycling performance and rate capability, showing a promising prospect for LIBs as anode materials.

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Compliance with ethical standards

Conflicts of interest The authors certify that they have no affiliations with or involvement in any organization or entity with any financial interest (such as honoraria; educational grants; participation in speakers' bureaus; membership, employment, consultancies, stock ownership, or other equity interest; and expert testimony or patent-licensing arrangements), or non-financial interest (such as personal or professional relationships, affiliations, knowledge, or beliefs) in the subject matter or materials discussed in this manuscript.

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